

# Towards Low-Coercive Field Operation of Sputtered Ferroelectric $\text{Sc}_x\text{Al}_{1-x}\text{N}$

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Primary CNF Tools Used: AJA sputter deposition, AJA ion mill, PT770, Oxford PECVD, SEM

## Abstract:

This work reports on the ferroelectric properties of scandium aluminum nitride ( $\text{Sc}_x\text{Al}_{1-x}\text{N}$ ) thin films with an Sc-concentration between 22-30%. The goal of this work is to engineer a low coercive field in  $\text{Sc}_x\text{Al}_{1-x}\text{N}$  for low-voltage post-CMOS compatible RF frontends. Films between 200 and 300 nm  $\text{Sc}_x\text{Al}_{1-x}\text{N}$  were deposited on platinum (Pt) and molybdenum (Mo) electrodes. Lateral and vertical capacitors were tested over a range of electric fields, frequencies, and electrode sizes. Measured coercive field ( $E_C$ ) and remnant polarization ( $P_r$ ) values were between 3.9-6.2 MV/cm and 58-170  $\mu\text{C}/\text{cm}^2$ . Frequency, temperature, and device area-dependence were studied to identify trends towards low coercive field operation of the material. An anomalous observation relating Sc-concentration and film stress, with  $E_C$  and  $P_r$  is reported to present prospects for decoupled tuning knobs to engineer lowered  $E_C$  in  $\text{Sc}_x\text{Al}_{1-x}\text{N}$ .

## Summary of Research:

Aluminum nitride (AlN) is the material of choice for MEMS RF resonators due to its excellent piezoelectric properties and CMOS compatibility. Fundamental limits in the piezoelectric properties of AlN have inspired the exploration of alloys of AlN towards enhanced piezoelectric properties. One such alloy is scandium aluminum nitride ( $\text{Sc}_x\text{Al}_{1-x}\text{N}$ ), which presents a tuning mechanism to increase the piezoelectric coefficient by  $\sim 4x$  with scandium incorporation. Theoretical calculations predict that monotonically increasing Sc-concentration can decrease the energy barrier between the parent wurtzite structure and the hexagonal phase in  $\text{Sc}_x\text{Al}_{1-x}\text{N}$  to produce ferroelectric switching. The recent discovery of ferroelectric switching in high Sc-concentration  $\text{Sc}_x\text{Al}_{1-x}\text{N}$  ( $x > 0.27$ ), has confirmed these predictions and generated significant interest as the first III-V ferroelectric [1]. Ferroelectricity has been demonstrated with Sc-concentration as low as 10%, and  $\text{Sc}_x\text{Al}_{1-x}\text{N}$  has been used for ferroelectric resonators [2,3]. A major challenge to use  $\text{Sc}_x\text{Al}_{1-x}\text{N}$  for device applications is the requirement for high on-chip voltages ( $>100\text{V}$ ) due to its high  $E_C$ . Here, we present results that map the design space of parameters which present a pathway towards lower  $E_C$  in  $\text{Sc}_x\text{Al}_{1-x}\text{N}$  [4,5].

**Fabrication.** Multiple film stacks were made for testing the impact of bottom electrode and substrate on  $\text{Sc}_x\text{Al}_{1-x}\text{N}$ . Ti/Pt was deposited on blanket silicon in the AJA sputtering

tool followed by  $\text{Sc}_x\text{Al}_{1-x}\text{N}/\text{Mo}$  deposition at an external vendor (OEM). In another variation,  $\text{SiO}_2$  was deposited in the Oxford PECVD followed by  $\text{Mo}/\text{Sc}_x\text{Al}_{1-x}\text{N}/\text{Mo}$  deposition at OEM. The film stacks were deposited in a cluster-line tool followed by etching of the top Mo and  $\text{Sc}_x\text{Al}_{1-x}\text{N}$  in the PT770 etcher. For the bottom electrode, Ti/Pt was etched in the AJA ion mill whereas the Mo was etched in the PT770. For etch-calibration, AlN films were deposited in the OEM Endeavor M1 and etched in the PT770.

**Results.** A setup with continuous wave positive-up-negative-down (PUND) was used for ferroelectric testing of  $\text{Sc}_x\text{Al}_{1-x}\text{N}$  (Figure 1A). The P and N pulses include both switching and leakage currents. The U and D pulses include only the leakage and can be subtracted from P and N pulses respectively to get the switching current (Figure 1B). Representative polarization vs. E-field (PE) loops are shown in Figure 1C.

*In situ* heating of capacitors on 300 nm  $\text{Sc}_{0.30}\text{Al}_{0.70}\text{N}$  shows a monotonic decrease in  $E_C$  without a significant change in  $P_r$  (Figure 2A). The temperature coefficient of  $E_C$  for this sample is  $\sim 8.8 \text{ kV}/^\circ\text{C}$  (Figure 2B). Figure 3 shows the area and frequency-dependence of  $E_C$  and  $P_r$  across a range of electrode sizes and frequencies from 20-100  $\mu\text{m}$  diameter and 1-8 kHz.  $E_C$  increases with an increase in

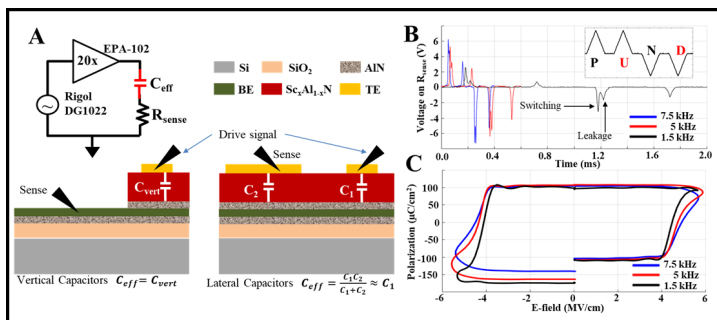


Figure 1: A) Modified Sawyer-Tower circuit for testing ferroelectric ScAlN capacitors and probing configurations for testing vertical and lateral configurations. B) Raw output voltage waveforms at various switching frequencies for PUND test (generic input waveform inset) C) Polarization vs. E-field (PE) loops generated by integrating switching currents from B.

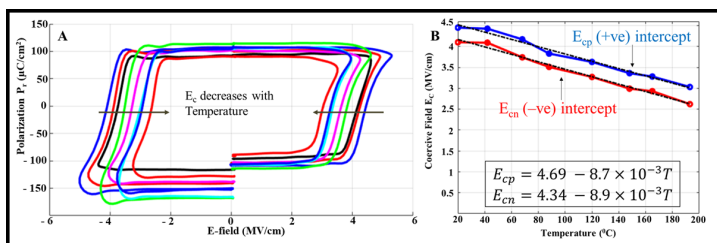


Figure 2: A) Polarization vs. E-field (PE) loops for 40 μm capacitor on 300 nm Sc<sub>0.30</sub>Al<sub>0.70</sub>N between RT and 195°C. B) Linear decrease in  $E_c$  with increase in temperature with a coefficient of  $\sim 8.8$  kV/cm/°C.

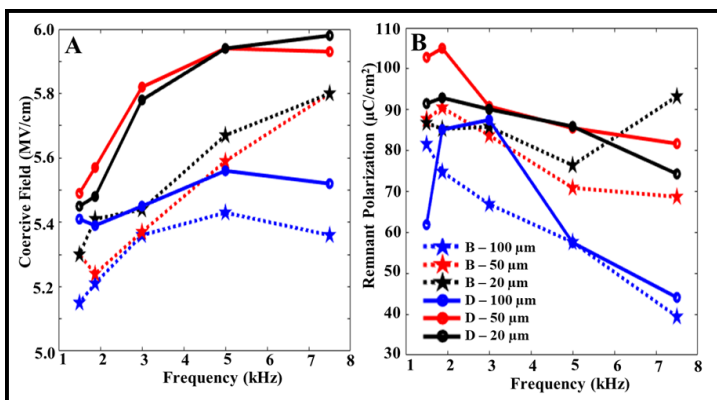


Figure 3: Plots of A)  $E_c$  and B)  $P_r$  versus frequency for capacitors of 20-100 μm diameter with Sc-concentration of 30% and 22% on Mo and Pt electrodes to map frequency and area-dependence of ferroelectricity.

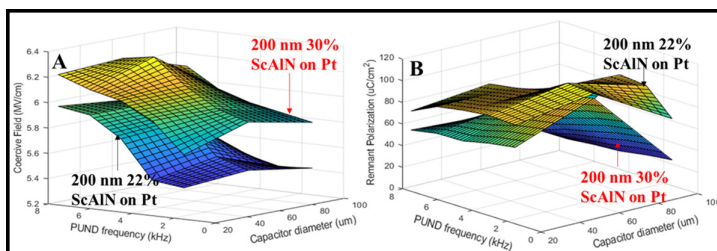


Figure 4: 3D map of A)  $E_c$  and B)  $P_r$  for 30% and 22% ScAlN on Pt electrodes tested over the range of capacitor sizes and frequencies. The 22% ScAlN has lower  $E_c$  and higher  $P_r$  due to a favorable tensile film stress of 102 MPa.

frequency and decreases with an increase in electrode size because a larger capacitor has more domains for ferroelectric switching and a lower frequency allows for more complete switching.  $P_r$  decreases with an increase in electrode size due to higher leakage in larger devices.

Figure 4 shows an anomalous dependence of  $E_c$  on Sc-concentration across a range of electrode sizes and frequencies — a lower Sc-concentration shows lower  $E_c$  due to a favorable tensile stress of 102 MPa compared to a compressive stress of 80 MPa. This stress-control shows a path towards decoupled knobs for tuning ferroelectricity, other than Sc-concentration.

## Conclusions and Future Steps:

Our work maps out the design space of low  $E_c$ -operation (as low as 2.5 MV/cm) in Sc<sub>x</sub>Al<sub>1-x</sub>N with *in situ* heating, and high frequency operation.

The use of this low  $E_c$  in ferroelectric transistors, resonators, transducers, and NEMS devices is the subject of on-going research.

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